

Self-Etching Synthesis of Superhydrophilic Iron-Rich Defect Heterostructure-Integrated Catalyst with Fast Oxygen Evolution Kinetics for Large-Current Water Splitting

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Developing catalysts with rich metal defects, strong hydrophilicity, and extensive grain boundaries is crucial for enhancing the kinetics of electrocatalytic water oxidation and facilitating large-current water splitting. In this study, we utilized pH-controlled etching and gas-phase phosphating to synthesize a flower-like Ni₂P-FeP₄-Cu₃P modified nickel foam heterostructure catalyst. This catalyst features pronounced hydrophilicity and a high concentration of Fe defects. It exhibits low overpotentials of 156 mV and 210 mV at current densities of 10 and 100 mA cm⁻² respectively, and maintains stability for up to 200 h at 100 mA cm⁻² with only 7.3% degradation, showcasing outstanding electrocatalytic water oxidation performance. Fur-

thermore, when integrated into a Ni₂P-FeP₄-Cu₃P/NF | Pt-C/NF electrolyzer, it achieves excellent overall water splitting performance, reaching current densities of 10 and 400 mA cm⁻² at just 1.47 V and 1.73 V, respectively, and operates stably for 60 h at 500 mA cm⁻² with minimal degradation. Analysis indicates that high-valence oxyhydroxides/phosphides of Ni, Fe, and Cu act as the primary active species. The presence of abundant Fe defects enhances electron transfer, strong hydrophilicity improves electrolyte contact, and numerous grain boundaries synergistically modulate the activation energy between active sites and oxygen-containing intermediates, significantly improving the kinetics of water oxidation.

Introduction

To advance the objective of sustainable development, exploring clean and sustainable energy sources as alternatives to fossil fuels is crucial.^[1] Hydrogen energy, recognized for its high energy density and environmental benefits, emerges as a compelling substitute.^[3] Electrochemical water splitting is a viable method for hydrogen production, yet its efficiency is hampered by the anodic oxygen evolution reaction (OER). This reaction on the anode is inherently slow, involving complex steps like the formation of the O=O bond and multiple proton-coupled electron transfers, both of which present high energy barriers.^[4] Despite the widespread use of Ir- and Ru-based materials in OER electrocatalysts, their scarcity, high cost, low current density, and limited durability present significant

challenges.^[6] Consequently, research efforts have increasingly focused on developing efficient, cost-effective OER electrocatalysts that can achieve high current density (> 400 mA cm⁻²) while offering enhanced durability.^[7]

In recent years, transition-metal phosphides (TMPs) have garnered attention due to their high catalytic activity, adjustable structure, stability, electrical conductivity, and composition.^[8] A popular strategy to boost TMPs' intrinsic activity involves incorporating heteroatoms to modify their electronic structure.^[10] Bimetallic or polymetallic phosphides often demonstrate enhanced electrochemical activity due to a synergistic effect among components. For example, Fe_{0.29}Co_{0.71}P nanosheet arrays on nickel foam have shown outstanding OER performance as anodes for overall water splitting (OWS), thanks to an electron density modulation strategy.^[12] Recent developments in heterostructured bimetallic phosphide, such as (Ni_{0.33}Fe_{0.67})₂P,^[13] FeP/Ni₂P,^[14] Ni₂P-Cu₃P,^[15] Fe-CoP,^[16] FeCo-P,^[17] and NiCoP,^[18] highlight their structural and performance benefits.

With insights into the OER process, TMP generally undergo dynamic self-reconstruction under oxidizing potentials to form metal oxides/(oxy)hydroxides, which exhibit significantly enhanced OER activity compared with directly synthesized oxide/(oxy)hydroxide counterparts.^[19] Despite the advancement in TMP-based electrocatalysts for OER, their activity is still insufficient for industrial needs. Further improvement in catalytic performance can be achieved by modifying their morphological characteristics.^[21] Grain boundary is a crystal defects that can introduce vacancies, dislocations, COFs-based metal-free heterojunctions^[24] and bond distortions, affecting the coordination environment of nearby atoms and altering the

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electronic structure of the active site. Integrating porous grain boundary catalysts with conductive substrates can adjust the electronic structure and create multiple new active sites, substantially boosting catalytic efficiency.^[25] Moreover, vacancy defects have been proven beneficial in creating active sites and enhancing electrocatalytic activities, although their higher formation energy poses a challenge compared to anion vacancies.^[27] Besides, the rate at which bubbles detach from the electrode surface and the electrode/electrolyte interface influences the mass transfer kinetics of generated gases and the catalytic efficiency of the catalyst. Hence, incorporating superhydrophilicity into the design of electrocatalysts represents an effective strategy for enhancing electrocatalytic performance.^[29] The strategic engineering of multi-phase grain boundary in TMPs, combining various morphologies, grain boundary, vacancy defects, and surface hydrophilicity properties to enhance intrinsic activity, remains highly desirable.^[30] Significant progress has been made in the development of electrolytic water-splitting catalysts over the past few decades.^[31] However, most of these catalysts are tested under controlled laboratory conditions, typically at current densities below 100 mA cm^{-2} .^[32] In contrast, industrial applications demand current densities ranging from 200 to 500 mA cm^{-2} . Therefore, it is crucial to focus research on designing catalysts that can perform effectively under high-current conditions.^[33]

Herein, we present a flower-like Fe defect-rich $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ catalyst developed through pH regulation and gas-

phase phosphorization, which displays superhydrophilicity and abundant grain boundaries. Characterization confirmed its flower-like structure with a larger specific surface area, potentially exposing more active sites. Superhydrophilicity was verified by contact angle tests, abundant Fe defects by electron paramagnetic resonance, and numerous grain boundaries by transmission electron microscopy. Electrochemical assessments showed that this catalyst requires only 156 and 210 mV overpotentials to achieve current densities of 10 and 100 mA cm^{-2} , respectively, in a 1.0 M KOH electrolyte. Furthermore, it achieved a cell voltage of 1.47 V at 10 mA cm^{-2} during overall water splitting, surpassing the benchmark $\text{RuO}_2 \parallel \text{Pt/C}$ (1.53 V), and maintained stability for over 60 h at 500 mA cm^{-2} . In situ characterization reveals that partial metal phosphides convert to hydroxide oxides at higher potentials, which, along with residual metal phosphides, serve as active species to lower the activation energy of oxygen-containing intermediates, thereby enhancing water oxidation kinetics.

Results and Discussion

Synthesis and Characterizations

The synthesis of flower-like $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ catalyst is depicted in Figure 1a. This study employed a pH-controlled one-step hydrothermal method to synthesize NiFeCu/NF pre-

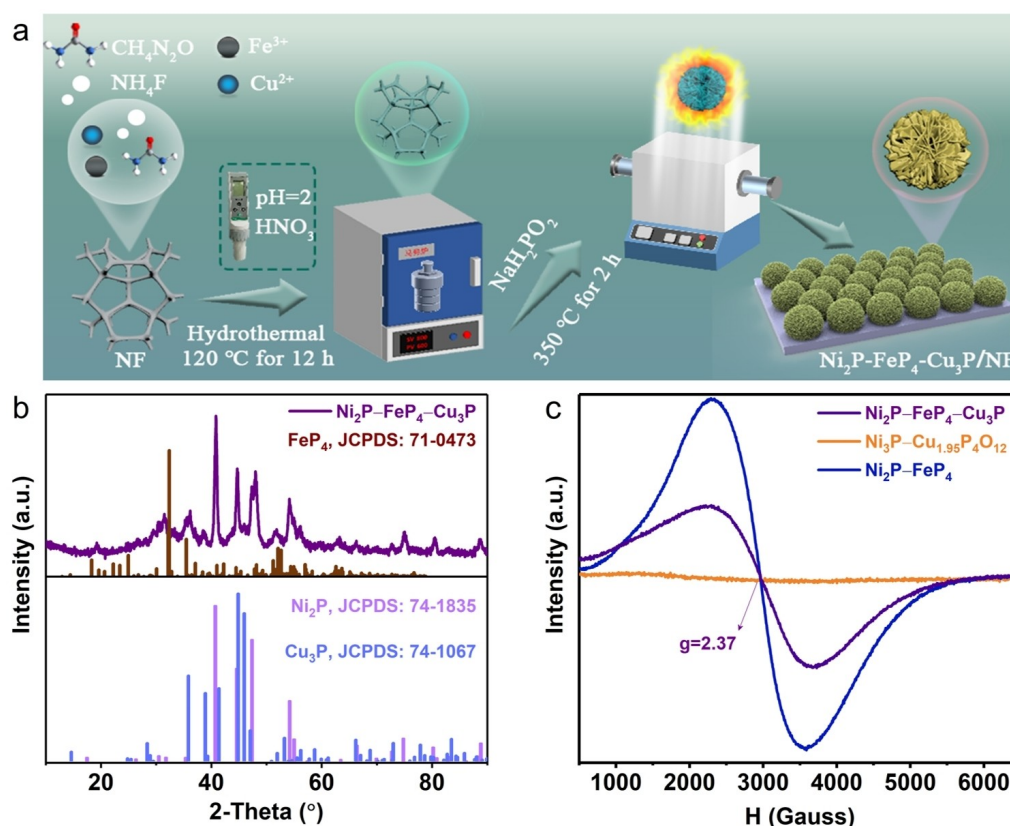


Figure 1. (a) Schematic illustration of the preparation of the $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$. (b) XRD pattern of $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$. (c) EPR spectra of $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$, $\text{Ni}_2\text{P-FeP}_4\text{/NF}$, and $\text{Ni}_3\text{P-Cu}_{1.95}\text{P}_4\text{O}_{12}\text{/NF}$.

cursor, which was subsequently transformed through gas-phase phosphorization to create a flower-like $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ target catalyst. The crystal structure of the catalyst was examined using X-ray diffraction (XRD). The XRD pattern shows that the $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ catalyst comprises crystalline phases of FeP_4 (JCPDS: 71–0473), Ni_2P (JCPDS: 74–1835), and Cu_3P (JCPDS: 74–1067) as illustrated in Figure 1b. In comparison, the precursor compounds of NF loaded with only Fe and Cu underwent phosphatization and transformed into $\text{Ni}_2\text{P-FeP}_4\text{/NF}$ and $\text{Ni}_3\text{P-Cu}_{1.95}\text{P}_4\text{O}_{12}\text{/NF}$, respectively (Figure S2a and b). The Raman spectra of $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$, shown in Figure S3, reveal four significant peaks at 180, 417, 463, and 1636 cm^{-1} , which correspond to the vibrational modes of Cu_2O ,^[35] NiO , $\alpha\text{-Ni(OH)}_2$, and O–H bending mode of H_2O ,^[36] respectively. Additionally, the type and concentration of defects were assessed by analyzing horizontal and vertical differences in the electron paramagnetic resonance (EPR) spectra. In the EPR spectra, both $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ and $\text{Ni}_2\text{P-FeP}_4\text{/NF}$ exhibited a similar signal around $g=2.37$ (Figure 1c), attributed to electrons trapped in Fe defects.^[38] $\text{Ni}_2\text{P-FeP}_4\text{/NF}$ showed the highest signal intensity, indicating a greater concentration of defects.^[39] It is important to note that the presence of moderate Fe defects in the target catalyst is critical for facilitating electron transfer and enhancing the reaction kinetics.^[41]

To examine the morphology of the prepared sample, both scanning electron microscopy (SEM) and transmission electron microscopy (TEM) were utilized. As shown in Figures 2a and S4, the $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$ retains its initial flower-like structure post-phosphorization, suggesting that the gas-phase phosphating process does not alter the morphology of the $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P}$ formed in situ on the NF surface. In contrast, $\text{Ni}_3\text{P-Cu}_{1.95}\text{P}_4\text{O}_{12}\text{/NF}$ exhibits a distinct blocky-shaped particle formation, densely distributed and anchored on NF (Figure S5a). $\text{Ni}_2\text{P-FeP}_4\text{/NF}$ presents an inhomogeneous structure that includes nanosheets and a flower-like formation (Figure S5b), whereas $\text{Ni}_2\text{P/NF}$ consists of nanorods (Figure S5c). The microsheet structures of $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-X}$ ($X=0.2, 0.4, 0.8,$ and 1.0) synthesized using solutions with varying concentrations of $\text{Cu(NO}_3)_2$ ($0.2, 0.4, 0.8,$ and 1.0 mmol , respectively) exhibit subtle differences (Figure S6a–d). It is also noteworthy that the microsheet structures of $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-pH}$ ($\text{pH}=1, 3, 4,$ and 5) display distinct variations (Figure S7a–d). As the pH of the solution increases, notably beyond a threshold of 3, the thickness of the nanosheet layers also increases. These morphological changes arise from variations in $\text{Cu(NO}_3)_2$ concentration and pH, which lead to increased copper ion concentrations in the solution. This, in turn, impacts the chemical and physical behavior of copper within the catalyst, causing changes in

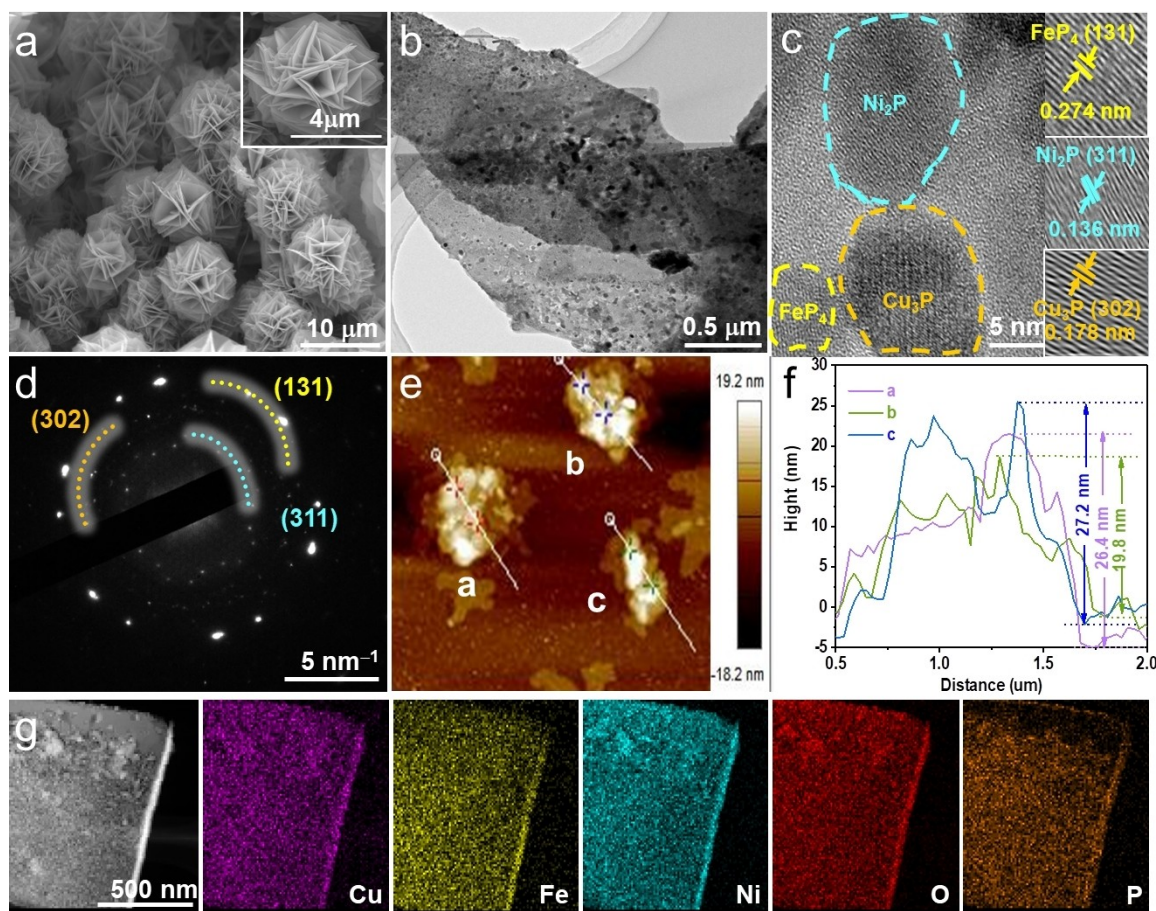


Figure 2. (a) SEM image, (b) TEM image, (c) HRTEM image, (d) SAED pattern, (e) AFM image, (f) corresponding height profiles and (g) HAADFSTEM image with corresponding elemental mappings of the $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$.

crystallinity, phase, morphology, pore structure, and surface elemental states. These alterations collectively influence both the morphology and performance of the catalyst.^[42] The optimal synthesis conditions were found using a solution with a pH of 2 and a $\text{Cu}(\text{NO}_3)_2$ concentration of 0.6 mmol, which not only improved the structural complexity but also significantly enhanced the number of active sites on the catalyst. The TEM image confirms that there are abundant particulate materials on the surface of the nanosheets (Figure 2b), and this rough surface facilitates the presentation of more active sites. Atomic force microscopy (AFM) measurements, as depicted in Figure 2e and f, reveal that the average thickness of the $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ nanoflake layer is approximately 24.4 nm. The high-resolution TEM (HRTEM) image in Figure 2c shows numerous dispersed amorphous areas alongside crystalline regions with varying orientations, highlighting a coexistence of crystalline and amorphous heterointerfaces. In these crystalline regions, Ni_2P , FeP_4 , and Cu_3P coexist, identified by lattice spacings of 0.136 nm (Ni_2P (311) plane), 0.274 nm (FeP_4 (131) plane), and 0.178 nm (Cu_3P (302) plane), suggesting that these phases are in close contact, forming distinct heterojunction interfaces that usually expose numerous active centers. The multiphase nature of the flower-like structure is further verified by selected area electron diffraction (SAED, Figure 2d).^[46] Additionally, the energy-dispersive X-ray (EDX) spectrum (Figure S8) and elemental mapping images (Figure 2g) confirm a uniform distribution of Cu, Fe, Ni, O, and P elements within the flower-like structure of $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$.

To explore the surface chemical composition and valence states of different catalysts, X-ray photoelectron spectroscopy

(XPS) was performed. The XPS survey spectra of $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ revealed distinct signals corresponding to the Ni, P, Cu, and Fe elements (red curve in Figure S9), aligning with XRD and EDX analysis results. The high-resolution C 1s spectrum (Figure S10) served as a reference for correcting other elements. In the high-resolution XPS spectra of Ni 2p_{3/2} (Figure 3a), peaks at 853.3 eV correspond to the Ni–P bond in both $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ and $\text{Ni}_2\text{P}-\text{FeP}_4/\text{NF}$. Additional peaks at 856.6 and 858.3 eV are attributed to Ni^{2+} and Ni^{3+} , respectively, with satellite peaks at 860.9 and 863.5 eV.^[47] The Ni–P bond in $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ exhibits a positive shift of 0.5 eV compared to $\text{Ni}_3\text{P}-\text{Cu}_{1.95}\text{P}_4\text{O}_{12}/\text{NF}$, indicating a strong electronic interaction between Fe and Ni species.^[20] Figure 3b shows that peaks at 706.9 eV primarily originate from the Fe–P bond in $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$, with $\text{Fe}^{2+}/\text{Fe}^{3+}$ peaks and a satellite peak observed at 710.9, 714.1 eV, and 717.7 eV, respectively.^[49] Compared to $\text{Ni}_2\text{P}-\text{FeP}_4/\text{NF}$, the Fe–P bond in $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ shifts towards a higher binding energy by 0.30 eV.^[20, 23] This shift indicates increased oxidation states of Ni and Fe cations, enhancing oxygen evolution reaction (OER) performance. The enhancement results from the doping of electronegative Cu, which modifies the electronegativity of P anions, increasing electron attraction to Ni and Fe, thereby raising their oxidation states and altering the surrounding electron density.^[50] Additionally, a lower $\text{Fe}^{2+}/\text{Fe}^{3+}$ peak area ratio (1.09) in $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ compared to $\text{Ni}_2\text{P}-\text{FeP}_4/\text{NF}$ (1.17) suggests a high density of Fe defects in the target catalyst.^[40, 52] The deconvoluted Cu 2p spectrum shown in Figure 3c exhibits peaks at 932.8/952.5 eV, attributed to the Cu 2p_{3/2} and Cu 2p_{1/2} states of Cu^+ , respectively, and peaks at 934.9/954.6 eV for Cu^{2+} . Two satellite peaks at 942.0/963.3 eV

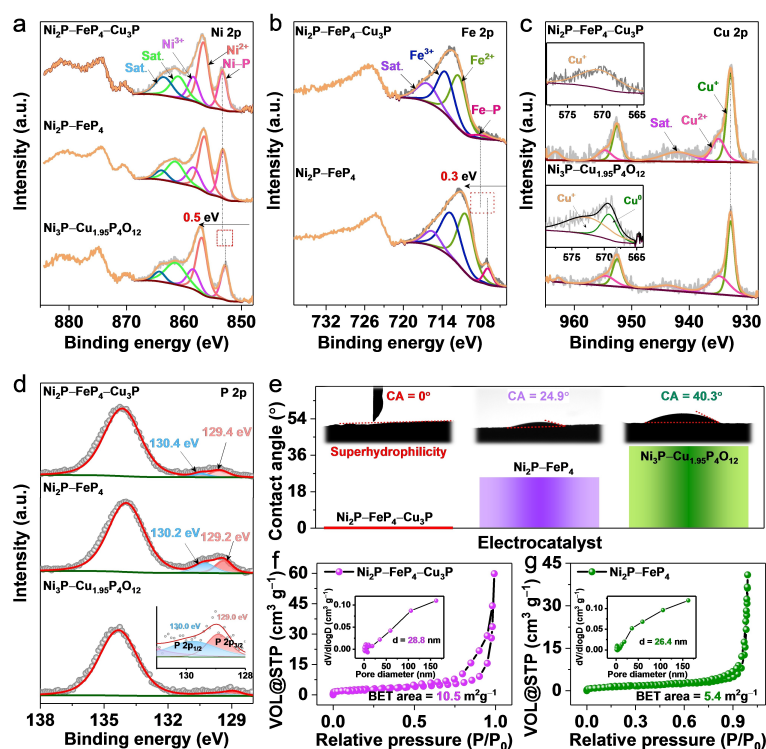


Figure 3. High resolution XPS spectra of (a) Ni 2p, (b) Fe 2p, (c) Cu 2p, and (d) P 2p of designed catalysts. (e) Contact angles of comparable catalysts. N_2 adsorption/desorption isotherms and pore size distributions (inset) of (f) $\text{Ni}_2\text{P}-\text{FeP}_4-\text{Cu}_3\text{P}/\text{NF}$ and (g) $\text{Ni}_2\text{P}-\text{FeP}_4/\text{NF}$.

correspond to Cu 2p_{3/2} and Cu 2p_{1/2}, respectively.^[53] A detailed analysis utilizing the Cu LMM Auger peak (inset of Figure 3c) identifies a prominent component at 570.4 eV, attributed to Cu⁺ species, confirming the presence of exclusively Cu⁺ in Ni₂P-FeP₄-Cu₃P/N. In contrast, the Ni₃P-Cu_{1.95}P₄O₁₂/NF catalyst exhibits two distinct valence states, Cu⁰/Cu⁺, aligning with XRD findings.^[56]

The high-resolution XPS spectra of P 2p in Ni₂P-FeP₄-Cu₃P/NF, shown in Figure 3d, reveal P 2p_{3/2} at 129.4 eV, P 2p_{1/2} at 130.4 eV, and P–O species at 134.1 eV on the surface. The binding energy of P–M (P 2p) bonds varies among catalysts, with the targeted catalyst exhibiting a slight positive shift, indicating strong interactions between metal atoms and P atoms.^[49, 57] This shift leads to enhanced local electric dipoles, which are beneficial for H₂O adsorption and weakening of the H–O bonds in adsorbed H₂O, thereby improving reaction kinetics.^[58]

The surface wettability of the fabricated electrodes was evaluated through contact angle measurements. The Ni₂P-FeP₄-Cu₃P/NF electrode shows superhydrophilic properties (Figure 3e), as water droplets immediately permeate the electrode surface, reducing the static contact angle to zero. In contrast, water contact angles on Ni₂P-FeP₄/NF and Ni₃P-Cu_{1.95}P₄O₁₂/NF were 24.9° and 40.3°, respectively. The flower-like structure of Ni₂P-FeP₄-Cu₃P/NF provides a high specific surface area that enhances its superhydrophilicity. This improvement is attributed to the increased surface roughness, which results from the dense distribution of small mesopores across its large surface area, thereby significantly improving surface wettability.^[59] This superhydrophilicity facilitates close electrolyte-electrode contact and rapid electrolyte diffusion, enhancing mass and charge transfer and accelerating reaction kinetics.^[14, 61] Brunauer-Emmett-Teller (BET) analysis of Ni₂P-FeP₄-Cu₃P/NF and Ni₂P-FeP₄/NF catalysts, shown in Figure 3f and g exhibits typical type III isotherms with mesoporous characteristics and distinct hysteresis loops.^[63] Ni₂P-FeP₄-Cu₃P/NF displays a higher BET surface area (10.5 m² g⁻¹) than Ni₂P-FeP₄/NF (5.4 m² g⁻¹), which helps expose more active sites, facilitates electrolyte diffusion, and improves gas emissions.^[65]

Electrocatalytic Performance

The OER performance of the catalysts was meticulously evaluated using a standard three-electrode setup in a 1.0 M KOH solution. All potentials were corrected for 100% iR and calibrated against the reversible hydrogen electrode (RHE, Figure S1). Notably, the Ni₂P-FeP₄-Cu₃P catalyst demonstrated exceptional catalytic performance under optimized conditions, achieved by adjusting the pH of the solution to 2 and the Cu(NO₃)₂ concentration to 0.6 mmol (Figures S12–S15). All subsequent discussions and analyses will focus on these optimal synthesis conditions.

As depicted in Figure 4a, the Ni₂P-FeP₄-Cu₃P/NF requires only 156 and 210 mV overpotentials to reach current densities of 10 and 100 mA cm⁻², respectively. These values are significantly lower than those required by Ni₃P-Cu_{1.95}P₄O₁₂/NF

(208 mV, 293 mV), Ni₂P-FeP₄/NF (184 mV, 241 mV), Ni₂P (160 mV, 338 mV), and RuO₂ (220 mV, 352 mV). Furthermore, the Ni₂P-FeP₄-Cu₃P/NF delivered an ultra-low overpotential of 211 mV at 100 mA cm⁻², outperforming all other tested electrocatalysts (Figure 4c). The OER performance of Ni₂P-FeP₄-Cu₃P/NF was also tested under various pH levels and Cu contents, with the findings indicating the lowest overpotential under the optimized conditions.

The superior OER performance is attributed to the unique flower-like structure of the catalyst, which facilitates electrolyte diffusion and enhances the synergistic effects of Ni, Fe, and Cu phosphorylation, thereby exposing numerous active sites.^[66] Tafel slopes calculated from linear sweep voltammetry (LSV) data indicate that Ni₂P-FeP₄-Cu₃P/NF has a Tafel slope of approximately 43.6 mV dec⁻¹ (Figure 4b), significantly lower than those of its counterparts. This suggests faster reaction dynamics than other catalysts such as Ni₃P-Cu_{1.95}P₄O₁₂/NF, Ni₂P-FeP₄/NF, and RuO₂ on nickel substrate. Ni₂P-FeP₄-Cu₃P/NF similarly outperforms recent non-precious metal catalysts in OER overpotential and Tafel slope, as shown in Figure 4d and Table S1. The electrochemical double-layer capacitance (C_{dl}) was measured through cyclic voltammetry (CV) in the non-Faradaic region at various scan rates (Figure S11).^[67] Ni₂P-FeP₄-Cu₃P/NF exhibited a C_{dl} of 134.0 mF cm⁻², significantly higher than those of Ni₃P-Cu_{1.95}P₄O₁₂/NF, Ni₂P-FeP₄/NF, Ni₂P/NF, and RuO₂/NF (Figure 4e). This high C_{dl} indicates an electrochemical surface area (ECSA) of 3350 cm² for Ni₂P-FeP₄-Cu₃P/NF, surpassing those of its competitors (Figure 4f), suggesting more active sites and enhanced catalytic performance.^[68] Nyquist plots further demonstrate minimal charge transfer resistance (R_{ct}) for Ni₂P-FeP₄-Cu₃P/NF compared to other catalysts, indicating rapid charge transfer across the electrode-electrolyte interface and faster kinetics for OER (Figure 4g, S12d and S14d).^[69] Durability tests also revealed robust long-term stability, with only a 7.3% decline in current density after 200 h at a constant current density of 100 mA cm⁻² (Figure 4h). The initial decline during the first 80 h, likely due to solution replenishment, was followed by gradual stabilization. Furthermore, it demonstrated robust stability during the OER stability test conducted at industrial-level current densities of 500 mA cm⁻². Further stability tests through multi-cycle CV indicated a negligible shift in polarization curves after 1000 and 7000 CV cycles (inset Figure 4h), underscoring the catalyst's durability. The slight decrease in catalytic performance is primarily attributed to catalyst aggregation, exfoliation, and changes in the surface chemical state (Figure 6c–f), emphasizing the robustness and potential practical applications of Ni₂P-FeP₄-Cu₃P/NF in OER processes.

Overall Water Splitting Performance

Leveraging the remarkable OER performance demonstrated by the as-synthesized Ni₂P-FeP₄-Cu₃P/NF in alkaline electrolytes, where it served as the anode paired with commercial Pt/C as the cathode, the overall water electrolysis performance was evaluated using a symmetrical two-electrode system (Figure 5a). To achieve a current density of 10 MA cm⁻² in 1.0 M KOH

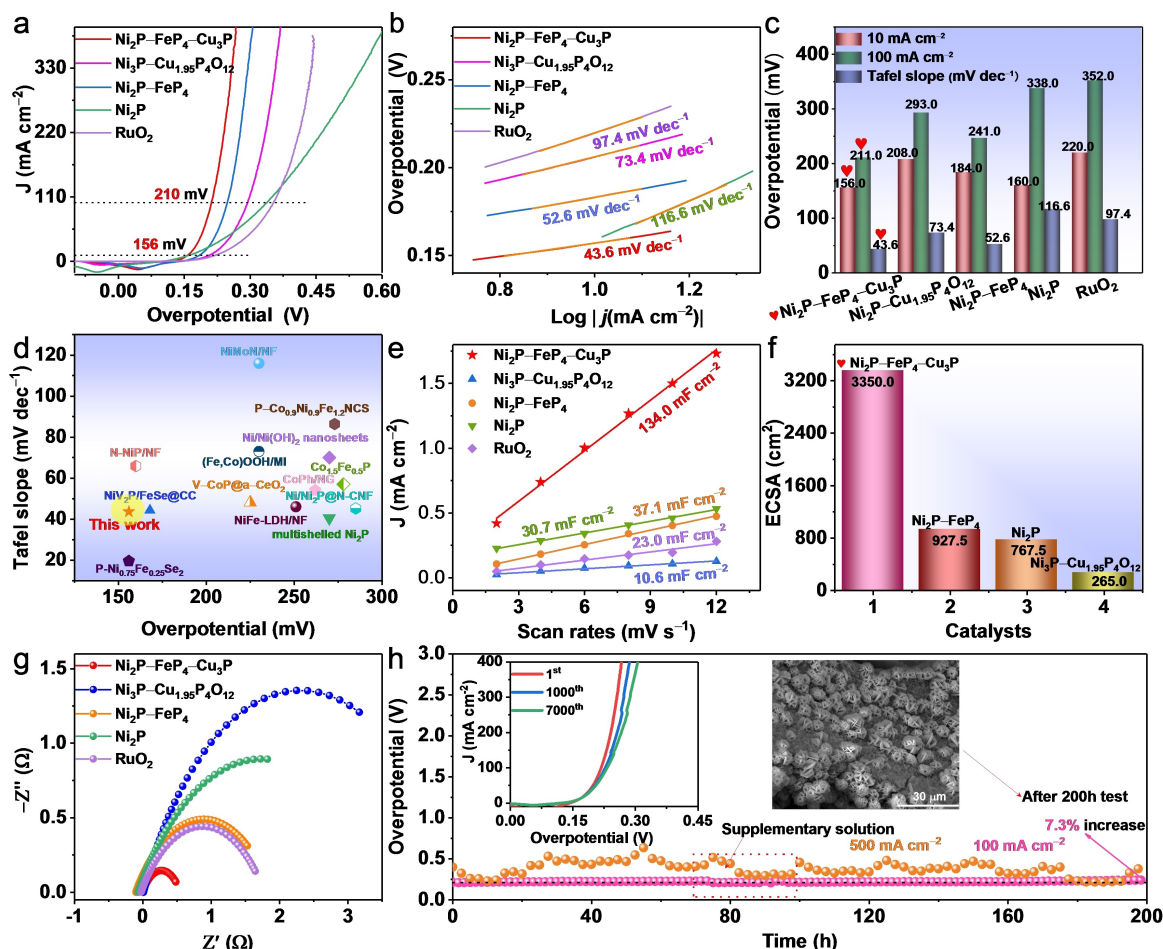


Figure 4. (a) LSV polarization curves, (b) Tafel slopes, and (c) summary of overpotentials (at 10 and 200 mA cm⁻²) and Tafel slopes of comparable catalysts in 1.0 M KOH. (d) Comparison of overpotential (at 10 mA cm⁻²) and Tafel slopes for various state-of-the-art OER catalysts. (e) C_{dl}, (f) summary of ECSA, and (g) electrochemical impedance spectroscopy of designed catalysts. (h) Stability of the Ni₂P-FeP₄-Cu₃P/NF catalyst for OER (the inset shows the corresponding LSV curves at different cycles; the image displays the morphology of the Ni₂P-FeP₄-Cu₃P/NF after stability).

electrolyte, the Ni₂P-FeP₄-Cu₃P/NF⁽⁺⁾ || Pt/C⁽⁻⁾ system requires a voltage of 1.47 V, which is notably lower than the 1.53 V required by the commercial RuO₂⁽⁺⁾ || Pt/C⁽⁻⁾ system (inset Figure 5b). As depicted in Figure 5b, the Ni₂P-FeP₄-Cu₃P/NF⁽⁺⁾ || Pt/C⁽⁻⁾ electrolyzer demonstrates exceptional efficiency, requiring only 1.73 and 1.85 V cell voltage to achieve high current densities of 400 and 800 mA cm⁻², respectively. This performance surpasses that of the commercial RuO₂⁽⁺⁾ || Pt/C⁽⁻⁾ electrolyzer, which requires 1.97 and 2.07 V to reach comparable current densities. Further comparisons are presented in Figure 5c and Table S2, showcasing the Ni₂P-FeP₄-Cu₃P/NF⁽⁺⁾ || Pt/C⁽⁻⁾ system's superior catalytic performance relative to other reported noble metal-free electrocatalysts. Owing to its hydrophilic surface properties and exceptional corrosion resistance, the Ni₂P-FeP₄-Cu₃P/NF⁽⁺⁾ || Pt/C⁽⁻⁾ system also demonstrated stable operation for more than 60 h (Figure 5d), even under industrial-level current densities of 500 mA cm⁻², albeit with some fluctuations. This stability and efficiency highlight its potential for practical applications in water electrolysis, particularly in settings requiring high durability and efficiency.

Dynamic Transformation Analysis

Electrochemical impedance spectroscopy (EIS) serves as an effective method for obtaining insights into the adsorption and desorption kinetics of reactants on the electrode surface.^[70] As shown in Figures 6a and S16a and b, the Nyquist plots from impedance measurements for OER process on Ni₂P-FeP₄-Cu₃P/NF, Ni₃P-Cu_{1.95}P₄O₁₂/NF, and Ni₂P-FeP₄/NF in 1.0 M KOH, ranging from an open circuit potential (OCP) to 1.50 V, reveal significant insights. The plots show large curved arcs in the low overpotential region, which notably shrink in the middle and high overpotential zones, suggesting enhanced kinetics and reactant adsorption as the potential increases. To better understand this behavior, the Nyquist plots are fitted with equivalent circuit diagrams (Figures S16 and S17), focusing on key parameters like the charge transfer resistance (R_{ct}) and the double-layer capacitance (C_{dl}). The R_{ct} represents the resistance against the adsorption of oxygen-containing reactants on the catalyst surface, while C_{dl} reflects charge transfer kinetics. Therefore, understanding the behavior of key oxo-species at the electrode/electrolyte interface through the analysis of adsorption

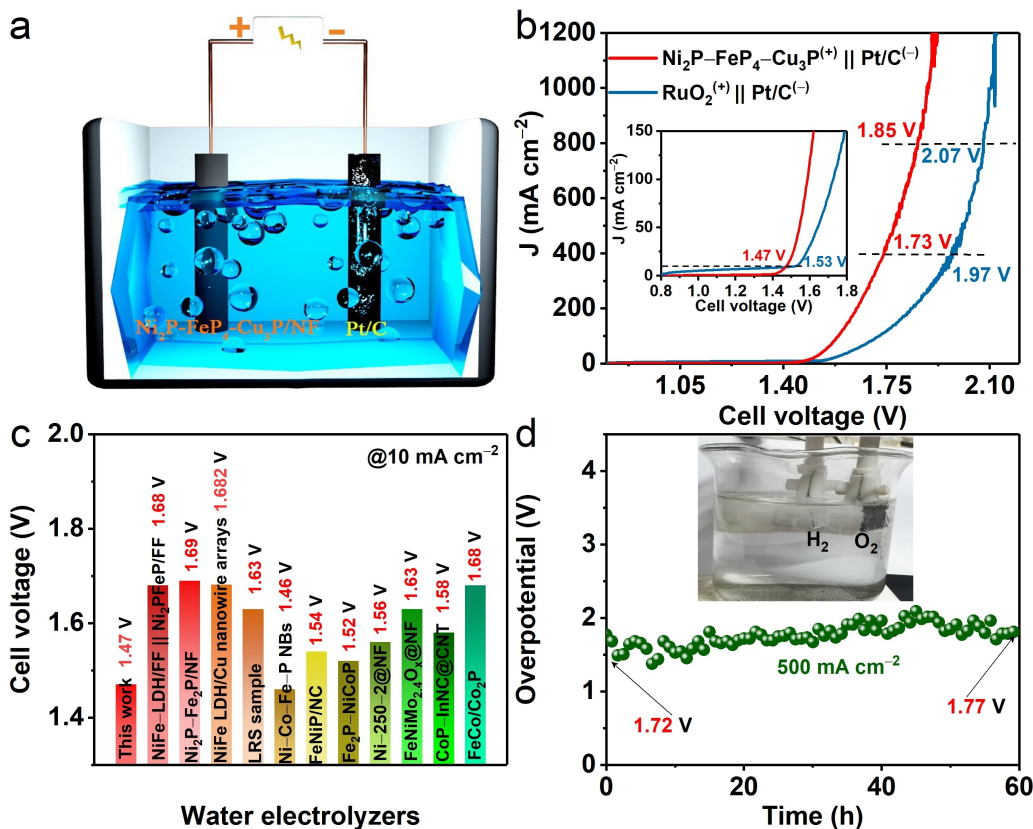


Figure 5. (a) Schematic diagram of overall water splitting in the two-electrode system. (b) LSV curves of overall water splitting for Ni₂P-FeP₄-Cu₃P/NF(+) || Pt/C(-) and RuO₂(+) || Pt/C(-) in 1.0 M KOH. (c) Comparison the cell voltage of Ni₂P-FeP₄-Cu₃P/NF at 10 mA cm⁻² with recently reported catalysts. (d) Chronopotentiometry test at 500 mA cm⁻².

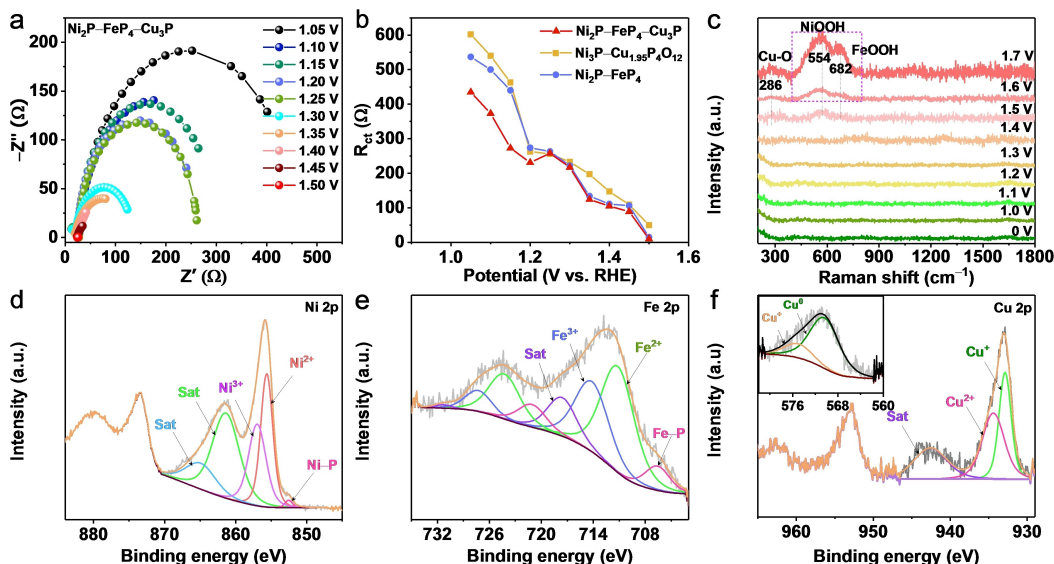


Figure 6. (a) Nyquist plots for the Ni₂P-FeP₄-Cu₃P/NF at different applied potentials in 1.0 M KOH. (b) Response of the ion adsorption-resistance (R_{ct}) at different potentials for various electrocatalysts. (c) Electrochemical in situ Raman spectra of the Ni₂P-FeP₄-Cu₃P/NF in the potential range of 0–1.70 V (vs. RHE). (d) Ni 2p, (e) Fe 2p, and (f) Cu 2p XPS spectra of the Ni₂P-FeP₄-Cu₃P/NF after OER stability (the inset shows Cu LMM Auger spectrum).

resistance (R_{ct}) is crucial for optimizing reaction processes. The alterations in R_{ct}, as depicted in Figure 6b, show that Ni₂P-FeP₄-Cu₃P/NF exhibits a lower R_{ct} compared to Ni₃P-Cu_{1.95}P₄O₁₂/NF and Ni₂P-FeP₄/NF, indicating faster adsorption kinetics of critical

oxygen-containing reactive intermediates during the OER process. The notably rapid decrease in R_{ct} for Ni₂P-FeP₄-Cu₃P/NF within the range of 1.1 V further underscores the swift adsorption of oxygen species at lower driving potentials,

resulting in the accumulation of key oxygen-containing reactive intermediates over the active hetero Ni₂P-FeP₄-Cu₃P/NF sites.^[72]

In-situ Raman spectroscopy was employed to meticulously monitor the structural evolution of the precatalysts and to investigate the nature of the reconstructed active species under OER conditions.^[73] In-situ Raman was used to track the transformation mechanism of Ni₂P-FeP₄-Cu₃P/NF during the OER process within the potential range of 1.0–1.7 V (vs. RHE, 1.0 M KOH), with a potential interval of 0.1 V (Figure 6c). At 0 V, the Raman spectrum represents the baseline data with no voltage applied. Upon increasing the potential to 1.4 V, two distinct peaks at 554 and 682 cm⁻¹ emerge, corresponding to NiOOH and δ-FeOOH, respectively, indicating their role as active centers during OER.^[74] Additionally, the peak at 286 cm⁻¹ related to Cu–O suggests oxidation of Cu during OER.^[77] These findings confirm that during the OER, Ni₂P-FeP₄-Cu₃P/NF undergoes surface reconstruction with in-situ formed NiOOH and δ-FeOOH acting as the active centers for OER.^[76, 78]

Further analysis of the high-resolution XPS spectra of Ni₂P-FeP₄-Cu₃P/NF post-OER stability (Figure 6d–f) revealed a significant increase in the area ratio of high-valence states of Ni³⁺, Fe³⁺, and Cu²⁺ (Ni³⁺/Ni²⁺ increased from 0.386 to 0.879, Fe³⁺/Fe²⁺ from 0.854 to 1.554, and Cu²⁺/Cu⁺ from 0.445 to 1.120). The XPS spectra of Ni₂P-FeP₄-Cu₃P/NF after the OER tests confirmed that the Ni–P, Fe–P, and Cu–P doublets were replaced by those corresponding to Ni–O, Fe–O, and Cu–O. This indicates that the true active species for the OER are Ni, Fe, and Cu in their oxidized forms. This analysis, coupled with the presence of P–M bonds in high-resolution P 2p (Figure S18) indicates that metal hydroxide oxides/phosphides are the main active sites during the OER.^[8, 13, 79]

The outstanding OER performance of the Ni₂P-FeP₄-Cu₃P/NF can be attributed to several key factors. Firstly, the flower-like structure provides a high specific surface area, which exposes more active sites and facilitates both electrolyte diffusion and gas release.^[80] Secondly, the superhydrophilicity of the catalyst ensures close contact between the electrolytes and active sites, accelerating charge transfer and enhancing reaction kinetics.^[49] Thirdly, the rich Fe defects in the Ni₂P-FeP₄-Cu₃P/NF catalyst improve conductivity and hydrophilicity, speeding up electron transfer and enhancing OER catalytic activity. Lastly, the synergistic effect of the abundant grain boundaries formed by the ternary metal phosphides optimizes the electronic structure of Ni₂P-FeP₄-Cu₃P/NF. In combination with the high-valence metal hydroxyl oxides generated during OER, these grain boundaries serve as true active sites, lowering the activation energy for oxygen-containing intermediates and boosting overall OER performance.^[82]

Conclusions

The Ni₂P-FeP₄-Cu₃P/NF catalyst has shown remarkable efficiency in OER, significantly outperforming traditional catalysts like RuO₂ by achieving lower overpotentials of 156 mV and 210 mV at current densities of 10 and 100 mA cm⁻², respectively. Kinetic studies reveal its rapid OER kinetics, evidenced by a Tafel slope

of 43.6 mV dec⁻¹, and electrochemical impedance spectroscopy indicates it has the lowest charge transfer resistance among tested materials. Long-term stability tests further demonstrate its industrial viability, with the catalyst maintaining high activity over 200 h under a constant current density of 500 mA cm⁻². In-situ Raman spectroscopy and post-OER X-ray photoelectron spectroscopy confirm structural resilience and the formation of active centers like NiOOH and δ-FeOOH under operational conditions. Moreover, the synergistic effect of Fe and Cu integration into the Ni₂P matrix enhances the overall catalytic activity, with high-valence metal oxides and phosphides identified as the primary active sites during OER. These attributes make Ni₂P-FeP₄-Cu₃P/NF a promising alternative to noble metal-based catalysts for water electrolysis applications, offering a blend of high efficiency, superior stability, and enhanced kinetic properties.

Experimental Sections

Materials

Iron (III) nitrate nonahydrate (Fe(NO₃)₃·9H₂O, 99%, Aladdin), copper (II) nitrate trihydrate (Cu(NO₃)₂·3H₂O, 99%, Aladdin), urea (CO(NH₂)₂, 99.0%, Aladdin), ammonium fluoride (NH₄F, 96.0%, Aladdin), sodium hypophosphite monohydrate (NaH₂PO₂·H₂O, 99%, Aladdin), nitric acid (HNO₃, 65%–68%, Xilong), potassium hydroxide (KOH, 90%, Macklin), ethanol (C₂H₅OH, 99.7%, Xilong), Nafion (5 wt%, Alfa Aesar), Pt/C (20 wt% Pt, Sinerco), nickel foam (NF) used as a substrate was obtained from Suzhou Sinerco Technology Co, Ltd (with a thickness of 1.6 mm). RuO₂ powder was prepared by directly annealing RuCl₃·3H₂O (37%, Inno-chem) at 400 °C in air. All chemicals were used as received without further purification.

Synthesis of NiFeCu/NF

A piece (1.5 cm×3.5 cm) of nickel foam (NF) was ultrasonicated with 0.5 M H₂SO₄, ethanol, and deionized water, respectively, for 15 min to remove impurities from the NF surface. Subsequently, 1.25 mmol Fe(NO₃)₃·9H₂O, 0.6 mmol Cu(NO₃)₂·3H₂O, 4 mmol NH₄F, and 4.5 mmol CO(NH₂)₂ were dissolved in 40 mL deionized water to form a uniform solution under magnetic agitation at room temperature. Then the HNO₃ aqueous solution was slowly added to adjust the pH value of the above-prepared solution (pH=2). After stirring the solution continuously for 30 min, the above treated NF was transferred together with the solution to a Teflon-lined stainless-steel autoclave (100 mL) and heated in an electric furnace at 120 °C for 12 h and then cooled naturally to room temperature. The obtained product (named NiFeCu/NF) was washed several times with ethanol and deionized water and then dried at 60 °C for 3 h.

NH₄F functions as an etching agent in catalyst synthesis, releasing active fluoride species (HF₂⁻ and NH₄⁺) through hydrolysis. These species etch metal surfaces and enhance catalytic performance. However, fluoride ions are typically removed due to multiple washings during synthesis, as indicated by the absence of F in EDX and XPS spectra.

Synthesis of Ni₂P-FeP₄-Cu₃P/NF

NaH₂PO₂·H₂O (1.0 g) and pre-prepared NiFeCu/NF were respectively placed upstream and downstream of the same tube furnace. The tube furnace was slowly heated to 350 °C for 120 min with a

heating rate of $5^{\circ}\text{Cmin}^{-1}$ in N_2 atmosphere to obtain $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$.

In addition, we limited the total amount of metallic Fe to 1.25 mmol and made a series of proportional adjustments by changing the amount of Cu metal. The synthesis method was the same for all variants, and the catalysts obtained were named $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P-0.2/NF}$, $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P-0.4/NF}$, $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P-0.8/NF}$, and $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P-1.0/NF}$. We synthesized other catalysts in the same way, differing only in the pH adjustment of the solution. The pH values were adjusted to pH=1, pH=3, pH=4, and pH=5, respectively. The catalysts obtained were labeled according to the pH of the solution: $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-pH=1}$, $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-pH=3}$, $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-pH=4}$, and $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF-pH=5}$.

In comparison, $\text{Ni}_2\text{P-FeP}_4\text{/NF}$, $\text{Ni}_3\text{P-Cu}_{1.95}\text{P}_4\text{O}_{12}\text{/NF}$, and NF-P were synthesized in a manner similar to $\text{Ni}_2\text{P-FeP}_4\text{-Cu}_3\text{P/NF}$.

Synthesis of $\text{RuO}_2\text{/NF}$ and 20 wt% Pt/C/NF Electrodes

RuO_2 (2 mg) and 20 wt% Pt/C were dispersed into a solution containing 200 μL deionized water, 200 μL ethanol and 10 μL of 5 wt% Nafion solution. The mixture was sonicated for 30 min to obtain a homogeneous solution. This solution was then drop-coated onto a clean NF (1 cm \times 1 cm) and dried in air.

Electrochemical Measurements

All electrochemical tests of the catalysts synthesized in this experiment were carried out on a Biologic VMP3 electrochemical workstation. The workstation uses a standard three-electrode system with an electrolyte solution of 1.0 M KOH (pH=13.5). The working electrode for electrochemical testing was a prepared NF-based catalyst (1 cm \times 1 cm), the counter electrode was a graphite plate, and the reference electrode was a standard calomel electrode (SCE). Linear sweep voltammetry, which was used to measure the electrochemical properties of the catalysts was reverse scanned and carried out in the range of 1.2–0 V (vs. SCE) at a sweep rate of 0.5 mVs^{-1} . Electrochemical impedance spectroscopy (EIS) measurements were determined at a voltage of 0.5 V (vs. SCE) over a frequency range of 300 kHz to 10 mHz, and finally the EIS data were fitted with ZSimDemo software. The electrochemical surface area (ECSA) was determined from the double-layer capacitance (C_{dl}), which in turn was obtained by cyclic voltammetry (CV) characterization. The CV characterization was measured in the non-Faraday interval voltage range of 0.1–0.2 V (vs. SCE) at scan rates of 2, 4, 6, 8, 10, and 12 mVs^{-1} respectively. The formula for calculating the C_{dl} was: $C_{dl} = (j_a - j_c) / (2 \times \nu)$, where j_a and j_c correspond to the anode and cathode current densities respectively, and ν represents the scanning rate. The ECSA was calculated as: $\text{ECSA} = C_{dl} / C_s$, where the C_s value was generally between 20 and 60 μFcm^{-2} , with an average of 40 μFcm^{-2} . The stability of the catalyst was measured by chronopotentiometry at a current density of 10 and 100 mAcm^{-2} for an extended time period. The overall water splitting tests were carried out in a 1.0 M KOH electrolyte with potentials ranging from 0–2.5 V and a scan rate of 5 mVs^{-1} . Experimentally measured potentials were converted to the RHE scale using the Nernst equation: E (vs. RHE) = $E_{\text{SCE}} + 0.059 \times \text{pH} + 0.0241\text{ V}$ and corrected with 100% iR according to equation.

In situ Raman spectroscopy experiments require a specific Raman spectrometer (InVia Qontor, Renishaw) equipped with an in situ test cell (Gaoss Union C031-1) for experimental determination in 1.0 M KOH solution. The as-prepared catalyst was used as the working electrode, the carbon rod as the counter electrode and the Ag/AgCl electrode as the reference electrode. The laser excitation wavelength was set at 785 nm and the exposure time for each spectral

recording was 10 min. The evolution of the catalyst was monitored by collecting Raman spectra at a constant potential of 1.0–1.7 V (vs. RHE).

In situ EIS measurements: In situ characterization with tandem electrochemical operation was carried out by maintaining the potential at 0.95–1.45 V (vs. RHE) for 10 min to obtain the surface chemical composition and structural information of the materials. EIS tests were performed at different potentials in the frequency range of 300 kHz to 10 mHz.

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Conflict of Interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data Availability Statement

Research data are not shared.

Keywords: pH-adjusted · Metal defects · Grain boundary · Oxygen evolution · Water splitting

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